# **Evaluation of Piezoelectric Ta<sub>2</sub>O<sub>5</sub> Thin Films Deposited on SrTiO<sub>3</sub> Substrates**

チタン酸ストロンチウム基板上への圧電性 Ta2O5 薄膜の成膜と評価

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# 1. Introduction

An X-axis-oriented tantalum pentoxide  $(Ta_2O_5)$ piezoelectric thin film has a strong piezoelectric property and a high dielectric constant.<sup>1</sup> Kakio, one of the authors, and colleagues found the optimum deposition conditions for obtaining a strong preferential (200) orientation and a high electromechanical coupling factor  $K^2$  for a Rayleigh-type surface acoustic wave (R-SAW) on a synthetic fused silica (SiO<sub>2</sub>) glass substrate. However, a large propagation loss (PL) for the R-SAW or bulk wave occurs in an oriented Ta<sub>2</sub>O<sub>5</sub> thin film.<sup>2,3</sup> By utilizing single-crystal Ta<sub>2</sub>O<sub>5</sub> thin films, a reduction in PL can be expected. Our group previously reported the fabrication of crystalline Ta<sub>2</sub>O<sub>5</sub> thin films on sapphire substrates by epitaxial growth.<sup>4</sup> However, the reduction of *PL* has not been confirmed.

In this study, strontium titanate (SrTiO<sub>3</sub>:STO) substrates were used for epitaxial growth because the lattice constant of STO is close to that of orthorhombic Ta<sub>2</sub>O<sub>5</sub> ( $\beta$ -Ta<sub>2</sub>O<sub>5</sub>), which has piezoelectricity.<sup>5</sup> The epitaxial growth of  $\beta$ -Ta<sub>2</sub>O<sub>5</sub> using an STO substrate was examined, and the crystallinity and R-SAW propagation properties of the samples were evaluated.

## 2. Sample Fabrication

 $Ta_2O_5$  thin films were deposited on STO(100) and STO(110) substrates using an RF magnetron sputtering system with a long-throw sputter cathode under the sputtering conditions shown in Table I. A SiO<sub>2</sub> substrate was also used for comparison. Figure 1 shows the process of sample fabrication. First, without applying O<sub>2</sub> radical power, a Ta<sub>2</sub>O<sub>5</sub> thin film having a thickness of approximately 0.3 µm was deposited as a buffer layer so that the  $Ta_2O_5$  thin film was not preferentially oriented along X-axis. Next, with the  $O_2$  radical power set to 150 W to obtain a piezoelectric Ta<sub>2</sub>O<sub>5</sub> thin film, a Ta<sub>2</sub>O<sub>5</sub> epitaxial thin film having a thickness of approximately 3.3 µm was deposited. The total film thickness h was 3.6 μm.

# 3. Evaluation of Crystallinity

First, the degree of orientation was evaluated

Table 1. Sputtering conditions for $1a_2O_5$ thin in	1lms.
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Gas ratio (Ar : O <sub>2</sub> )	30:10
Gas pressure [Pa]	0.75
Substrate heating [°C]	800
RF power (Ta) [W]	150
O <sub>2</sub> radical power [W]	$0 \rightarrow 150$
Total deposition time [h]	4.2
Ta <sub>2</sub> O <sub>3</sub> buffer layer	Ta <sub>2</sub> O <sub>5</sub> epitaxial thi Film thickness : 3.3



from X-ray diffraction (XRD) patterns using a CuK $\alpha$  X-ray source. Figure 2 shows the XRD patterns of a Ta<sub>2</sub>O<sub>5</sub>/STO(100) sample (A), a  $Ta_2O_5/STO(110)$  sample (B), and a  $Ta_2O_5/SiO_2$ sample (C). Compared with the diffraction intensity of the Ta<sub>2</sub>O<sub>5</sub>(200) plane of sample (C), those of samples (A) and (B) are about 6 and 25 times higher, respectively. Therefore, it was found that highly oriented Ta<sub>2</sub>O<sub>5</sub> thin films can be deposited on STO substrates. However, the diffraction angles  $2\theta$ corresponding to the (200) plane of samples (A) and (B) are different from that of the  $Ta_2O_5/SiO_2$ sample (C). The difference is considered to be due to the in-plane lattice arrangement, as mentioned later. Moreover, the Ta<sub>2</sub>O<sub>5</sub> thin film of sample (B) was observed to have two lattice plane spacings around the (200) plane. In addition, since the strongest peaks of the (200) plane of samples (A) and (B) appeared at a lower diffraction angle than that of the Ta<sub>2</sub>O<sub>5</sub>/SiO<sub>2</sub> sample (C), there is a possibility that the piezoelectricity of samples (A) and (B) is lower than that of the Ta<sub>2</sub>O<sub>5</sub>/SiO<sub>2</sub> sample.<sup>2</sup>

Next, the in-plane crystallinity of the  $Ta_2O_5$  thin films was evaluated by using a pole figure plot. **Figure 3** shows the measured pole figures of these samples. Poles corresponding to {201} were observed for samples (A) and (B). Regarding the symmetry of the poles, samples (A) and (B) have a fourfold symmetrical peak and a twofold symmetrical peak respectively, while sample (C) has a ring-shaped pattern. It is considered that sample

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(A) STO(100), (B) STO(110), and (C) SiO<sub>2</sub> samples. Fig. 3 Pole figures.

(A) has two lattice arrangements orthogonal to each other and sample (B) has a single arrangement. On the other hand, sample (C) has no in-plane lattice arrangement.

From the above results, the possibility of the homoepitaxial growth of  $\beta$ -Ta<sub>2</sub>O<sub>5</sub> on an STO substrate was demonstrated. In particular, it was found that a high-crystallinity Ta<sub>2</sub>O<sub>5</sub> thin film was deposited on the STO(110) substrate.

## 4. Evaluation of SAW properties

Interdigital transducers (IDTs) with a period  $\lambda$  of 8 µm and *N*=30 finger pairs were fabricated on the above samples by photolithography using an Al film, then the piezoelectricity and propagation properties of the Ta<sub>2</sub>O<sub>5</sub> thin films were evaluated by measuring the R-SAW propagation properties.

The frequency responses of these samples with  $h/\lambda$ =0.45 were measured using a network analyzer. **Figure 4** shows the frequency responses of samples (B) and (C), and **Figure 5** shows the minimum insertion loss *MIL* as a function of the propagation length *L* of these samples. The *PL* of the zeroth mode were measured from the *MIL* vs *L* relationships to be 0.11 dB/ $\lambda$  for sample (A), 0.02 dB/ $\lambda$  for sample (B), and 0.21 dB/ $\lambda$  for sample (C). Therefore, the Ta<sub>2</sub>O<sub>5</sub> thin films deposited on the STO substrates had a smaller *PL* than the Ta<sub>2</sub>O<sub>5</sub>/SiO<sub>2</sub> sample. However, the *MIL* values of samples (A) and (B) were larger than that of sample (C).

The coupling factor  $K^2$  was measured from the admittance of the IDTs. For samples (A) and (B), the



Fig. 5 Minimum insertion loss vs propagation length.

enhancement of  $K^2$  can be expected because  $\beta$ -Ta<sub>2</sub>O<sub>5</sub> has piezoelectricity. However, the  $K^2$  values of samples (A) and (B) were about 0.1%, while that of sample (C) was 0.87%. Therefore, although the crystallinity of the Ta<sub>2</sub>O<sub>5</sub> thin films of samples (A) and (B) was high, the piezoelectricity was low. The low piezoelectricity of the crystalline Ta<sub>2</sub>O<sub>5</sub> thin films on the STO substrates is considered to be due to the nonuniformity of the polarization direction.

### **5.** Conclusions

In this study,  $Ta_2O_5$  thin films were deposited on  $SrTiO_3(100)$ ,  $SrTiO_3(110)$ , and  $SiO_2$  substrates using an RF magnetron sputtering system, and the crystallinity and R-SAW propagation properties of the samples were evaluated. The possibility that the  $Ta_2O_5$  thin films were crystallized to orthorhombic  $Ta_2O_5$  with piezoelectricity on  $SrTiO_3$  substrates was demonstrated. The *PL* of  $Ta_2O_5$  thin films deposited on  $SrTiO_3$  substrates was maller than that of the film deposited on a  $SiO_2$  substrate; however, their piezoelectricity was lower than that of the  $Ta_2O_5/SiO_2$  sample. As the next step, we will apply a polarization process to  $Ta_2O_5$  thin films to enhance their piezoelectricity.

### References

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